Self-consistent Modeling of Polycrystal Plasticity

B. Clausen[†], C.N Tomé[‡], P. Maudlin^{*} and Mark Bourke[‡]

LANL, †LANSCE-12, ‡MST-8, *T-3

TMS 2000, Nashville, Tennessee

March 12-16, 2000



Polycrystal versus Continuum Constitutive Description





Outline

- Self-consistent model (SCM)
 - Input, assumptions, output
- Internal and residual strain development (EPSC)
 - Tensile testing of stainless steel
- Influence of deformation modes on texture development (VPSC)
 - Clock rolled Zircaloy
- Incorporating SCM into FEM formulation
 - Bending of highly textured zirconium bars
- Conclusions



Self-consistent model (SCM) - EPSC

- Material parameters
 - Single crystal stiffnesses and coefficients of thermal expansion
 - Description of texture with discrete set of grain orientations
 - Crystal structure, slip (and twinning) systems
 - CRSS and hardening law
- Model Assumptions
 - Eshelby inclusion theory
 - HEM properties equal to weighted average of the grains
- Output
 - Direct comparison with neutron diffraction measurements
 - Averages over grains sets representing reflections





Self-consistent model (SCM) - basic equations

$$\begin{split} \dot{\boldsymbol{\varepsilon}}_{c}^{P} &= \sum_{i} \boldsymbol{\mu}^{i} \dot{\gamma}^{i} & \dot{\boldsymbol{\varepsilon}}_{c}^{P} = \sum_{s} \boldsymbol{\mu}^{s} \dot{\gamma}^{s} \\ \sum_{j} \dot{\gamma}^{j} X^{ij} &= \boldsymbol{\mu}^{i} \boldsymbol{\mathcal{L}}_{c} \dot{\boldsymbol{\varepsilon}}_{c} , \quad X^{ij} = h^{ij} + \boldsymbol{\mu}^{i} \boldsymbol{\mathcal{L}}_{c} \boldsymbol{\mu}^{j} \\ \dot{\gamma}^{i} &= \boldsymbol{f}^{i} \dot{\boldsymbol{\varepsilon}}_{c} , \quad \boldsymbol{f}^{i} = \sum_{k} Y^{ik} \boldsymbol{\mathcal{L}}_{c} \boldsymbol{\mu}^{k}, \quad Y^{ij} = (X^{ij})^{-1} & \dot{\gamma}^{s} = \dot{\gamma}_{0} \sum_{s} \boldsymbol{\mu}^{i} \left(\frac{\boldsymbol{\mu}^{s} \cdot \boldsymbol{\sigma}'}{\tau_{c}^{s}} \right)^{n} \\ \boldsymbol{L}_{c} &= \boldsymbol{\mathcal{L}}_{c} \left(\boldsymbol{I} - \sum_{m} \boldsymbol{\mu}^{m} \boldsymbol{f}^{m} \right) & \boldsymbol{M}_{c} = \dot{\gamma}_{0} \sum_{s} \frac{\boldsymbol{\mu}_{s}^{i} \boldsymbol{\mu}_{s}^{j}}{\tau_{c}^{s}} \left(\frac{\boldsymbol{\mu}_{s}^{s} \boldsymbol{\sigma}_{s}'}{\tau_{c}^{s}} \right)^{n-1} \\ \dot{\boldsymbol{\varepsilon}}_{c} &= \boldsymbol{M}_{c} \dot{\boldsymbol{\sigma}}_{c}, \quad \boldsymbol{M}_{c} = (\boldsymbol{L}_{c})^{-1} & \dot{\boldsymbol{\varepsilon}}_{c} = \boldsymbol{M}_{c} : \boldsymbol{\sigma}_{c}' \end{split}$$

 $egin{aligned} oldsymbol{A}_c &= \left(oldsymbol{L}_c + ilde{oldsymbol{L}}
ight)^{-1} \left(oldsymbol{L} + ilde{oldsymbol{M}}
ight) &oldsymbol{B}_c &= \left(oldsymbol{M}_c + ilde{oldsymbol{M}}
ight)^{-1} \left(oldsymbol{M} + ilde{oldsymbol{M}}
ight) &oldsymbol{B}_c &= \left(oldsymbol{M}_c + ilde{oldsymbol{M}}
ight)^{-1} \left(oldsymbol{M} + ilde{oldsymbol{M}}
ight) &oldsymbol{M} &= \left\langleoldsymbol{B}_c oldsymbol{M}_c
ight\rangle &oldsymbol{M} &= \left\langleoldsymbol{B}_c oldsymbol{M}_c
ight\rangle &oldsymbol{M} &= \left\langleoldsymbol{B}_c oldsymbol{M}_c
ight
angle &oldsymbol{M} &= \left\langleoldsymbol{B}_c oldsymbol{M}_c
ight
angle &oldsymbol{M}_c
ight
angle &oldsymb$

EPSC (small strain)

VPSC (large deformations)

- Basic equations for the EPSC and VPSC formulations
- Solve last two sets of equations iteratively



Comparison to neutron diffraction data - EPSC



- Uniaxial tensile loading of austenitic stainless steel
- ND measurements made at load levels marked by the symbols
- Schematic set-up of the NPD at LANSCE. Measurement time is about 2-3 hours
- Measure elastic strains in two directions simultaneously



Comparison to neutron diffraction data - EPSC





- Applied stress versus measured elastic lattice strain
- ND measures lattice spacing changes and thereby only elastic strains
- Symbols are measurements, lines are model predictions

- The reflections carry different amount of the load
- Plasticity starts around the <531> orientation
 - => the <531> reflection deflects to the left
- The <100> orientation stays elastic the longest
 - => the <200> reflection deflects to the right



Self-consistent model (SCM) - VPSC



- Prediction of rolling textures for Zircaloy, rolled to a true strain of 1.0
- Two different model assumptions of plastic deformation modes
 - prismatic, tensile twins and compressive twins versus prismatic, tensile twins and pyramidal
- Comparison to measured textures for Zircaloy with two different grain sizes



SCM and FEM



- Highly textured "clock" rolled Zirconium
- One set of deformation modes, CRSS and hardening parameters
- Possible to reproduce measured material behavior for all three deformation tests
 - Through thickness compression (TTC), in-plane compression (IPC) and in-plane tension (IPT)



SCM and FEM



- 4-point bending of heavily textured Zirconium bars
- Difference in deformed cross-section depending on orientation of preferred c-axis orientation (C0 or C90)
- 377 grain orientations for each element, 4000 elements
- Good agreement between predicted cross-sections and measured data



Conclusions

- Self-consistent modeling (polycrystal constitutive model)
 - Very useful tool for interpreting neutron diffraction data
 - Pinpoint active deformation mechanisms
- Prediction of internal and residual stresses and strains (EPSC)
- Prediction of texture development for large strains (VPSC)
- Incorporated SCM into FEM formulation



Self-consistent model (SCM) - EPSC

- Known elastic properties
 - Calculate HEM stiffness using Voigt, Reuss or Hill average
 - Calculate stress and strain increments in all grains for given macroscopic stress or strain increment
 - Requires knowledge of stiffness of HEM and all the grains
 - Determines a new self-consistent average stiffness
 - Compare the new and old average stiffness/compliance
 - Iterate until sufficient convergence is obtained
- Further deformation
 - Yielding in grains (stress exceeds yield criteria)
 - Determine the elastic-plastic stiffness
- "Bookkeeping"
 - Calculate elastic strains for grain sets representing reflections
 - Statistics on; # of slip systems, Taylor factor, plastic strain, etc.



- Somewhat different length scale "meso scale"
- Self-consistent polycrystal models
- Tool to predict the macroscopic behavior of a material using microscopic material behavior, such as single X-tal elastic stiffness, inelastic deformation modes like slip and twining, and texture
- Show how we can validate this tool with neutron diffraction measurements of elastic lattice strains and texture measurements
- Start out with describing how we can use the SC models as a constitutive material formulation

